Role of Ag doping in Ba₈Si₄₆ compounds

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The silicon clathrate compound Ba₈Si₄₆ shows superconductivity below the critical temperature (T_c) of 8 K, and the T_c decreases monotonically with doping Ag. In order to reveal effects of Ag doping on the electronic states, we have applied soft x-ray photoemission spectroscopy to Ag-doped silicon clathrate compounds Ba₈Ag_xSi_{46-x} (x=0,1,3,6). The valence band photoemission spectra show that a Ba 5*d*-derived state at the Fermi level (E_F), which is prominently observed in Ba₈Si₄₆, decreases with increasing Ag content. The reduction in the peak intensity at E_F with increasing Ag content is therefore in accord with the decrease of T_c in Ba₈Ag_xSi_{46-x}. Band structure calculation using local-density approximation reproduces the observed valence band spectra of x=0 and 6. The Si 2p and Ba 4*d* core-level photoemission spectra demonstrate that the valence electron of Si is attracted to the Ag site in x=1 and the 5*d* electron of Ba inside the Si₂₄ cage is further donated to Ag in $x \ge 3$. Hence, Ag doping leads to the reduction of the peak at E_F .

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Studies on silicon clathrate compounds trace back to discussion on the metal-insulator transition in Na_xSi₄₆ and Na_xSi1_{36} .¹ Recently, $Na_2Ba_6Si_{46}$ and Ba_8Si_{46} have been found to show superconductivity with T_c of 4 and 8 K, respectively,²⁻⁹ which have led to much notice into Si clathrate compounds. In Ba₈Si₄₆, Si atoms form a network consisting of Si₂₄ and Si₂₀ cages and Ba atoms are located at the centers of these Si cages (Fig. 1). Ba₈Si₄₆ is similar to alkalimetal doped C₆₀ materials in terms of a polyhedra-network system showing superconductivity.^{10–15} In the C_{60} materials, the polyhedra are held together by van der Waals-type interactions. The superconductivity in the alkali-metal-doped C_{60} materials is based on the competition between the on-site Coulomb repulsion energy among the electrons occupying t_{1u} -derived C₆₀ bands and the coupling between these electrons and intramolecular vibrations of C₆₀. In Si clathrate compounds, however, the polyhedra share the pentagonal and hexagonal faces as shown in Fig. 1 and form a covalently bonded network. The band structure calculation by the density functional theory for Ba₈Si₄₆ has predicted that strong hybridization between the Si 3p and Ba 5d states modifies the density of states (DOS) of the Si network and forms a narrow peak at the E_F .^{7,16,17} This hybridized state at E_F is thought to play a crucial role in the superconductivity of Ba_8Si_{46} , since T_c is a function of DOS at E_F in normal state $[N(E_F)]$ through the coupling constant $\lambda = N(E_F)V$, where V is electron-excitation coupling strength.¹⁸ The photo emission spectroscopy using He I α (21.218 eV) and He II α (40.814 eV) has shown the narrow peak at E_F and has observed the superconducting gap $[2\Delta(E_F)/k_BT_c]$ of 4.38 in the superconducting state.¹⁹ The hybridized state at E_F has been also observed by the Knight shift in nuclear magnetic resonance (NMR)^{20,21} and the specific-heat and magnetic susceptibility measurements.⁵ On the other hand, the study by neutron scattering and extended x-ray absorption fine structure (EXAFS) has shown that the vibrational coupling between Ba and Si network inside the Si₂₄ cage is also the important parameter to explain consistently the superconductivity in Ba₈Si₄₆ and the absence of superconductivity in Na₈Si₄₆ for which $N(E_F) \neq 0.^{22}$ The recent study on the isotope effect in Raman spectrum has indicated that the superconductivity in Ba₈Si₄₆ is explained by Bardeen-Cooper-Schrieffer (BCS) mechanism with medium coupling regime.⁵

In Ba₈Ag_xSi_{46-x} studied here, Si atoms are replaced with Ag atoms at 6*c* sites which belong only to the Si₂₄ cages. The 6*c* sites are represented by black circles in Fig. 1 which shows the crystal structure of Ba₈Ag_xSi_{46-x}. In Ba₈Ag₃Si₄₃, Ag atoms occupy the 6*c* sites alternately in the Si sixfold ring. Ba₈Ag_xSi_{46-x} shows the superconductivity in $x \le 3$, while the T_c decreases with increasing Ag content to 7 K and 2.8 K for x=1 and 3, respectively. The superconductivity has not been observed down to 1.8 K in Ag-doped clathrates with x > 3.²³ Therefore, the systematic study for the elec-

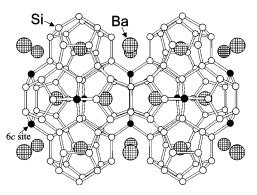


FIG. 1. Crystal structure of $Ba_8Ag_xSi_{46-x}$. Black circles indicate the 6*c* sites where Si atom is replaced with Ag atom by Ag doping.

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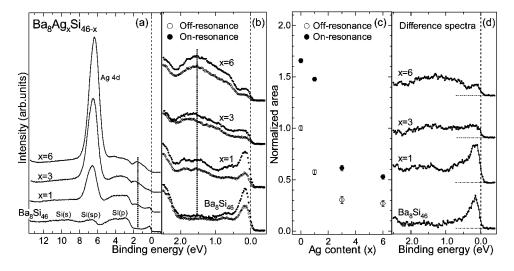


FIG. 2. (a) The valence band spectra of $Ba_8Ag_xSi_{46-x}$ (x=0,1,3,6), measured with $h\nu=780$ eV. The spectra are normalized by the intensity at 12.0 eV. (b) Resonant photoemission spectra at Ba 3*d* threshold. The closed and open circles show the on-resonant ($h\nu=789$ eV) and off-resonant ($h\nu=780$ eV) spectra of $Ba_8Ag_xSi_{46-x}$ (x=0,1,3,6). The spectra are normalized by the photon flux. (c) The areas under the curves between E_F and 0.35 eV of the on- and off-resonant spectra shown in (b) are plotted by the closed and open circles with error bars, respectively. All values are normalized by the value for the off-resonant spectrum of Ba_8Si_{46} . (d) The difference spectra between the on- and off-resonance spectra of $Ba_8Ag_xSi_{46-x}$ (x=0,1,3,6).

tronic states of $Ba_8Ag_xSi_{46-x}$ as a function of Ag content (x) provides the important knowledge on the superconductivity in Si clathrate compounds such as the mechanism of the change in the T_c and the appearance and absence of the superconductivity, which are indeed related to the behavior of the DOS at E_F . Here, in order to study the electronic states of $Ba_8Ag_xSi_{46-x}$ depending on Ag content (x) and to elucidate the change in them caused by Ag doping, we have applied soft x-ray photoemission to $Ba_8Ag_xSi_{46-x}$ with x=0, 1, 3, and 6. An advantage in the use of soft x-ray is a long photoelectron mean free path compared to the photoemission using ultraviolet rays and, hence, the effective probe of bulk electronic states. The valence band photoemission has probed the DOS at E_F depending on Ag content and has shown the additional bonding states at ~1.5 eV binding energy in Agdoped Si clathrates. In addition, the resonant photoemission at Ba 3d threshold, which enhances the Ba states selectively, has been performed to study the partial density of states (pDOS) of Ba in the valence band. Ba 4d and Si 2p corelevel photoemission have been also done to study the electronic states of $Ba_8Ag_xSi_{46-x}$ (x=0,1,3,6) with element specification. These core-level spectra elucidate the interchange of the charge between the elements and the origin of changes in the valence band.

 $Ba_8Ag_xSi_{46-x}$ (x=0,1,3,6) were prepared under high pressure.^{4,6,7,23} Ba, Si, and Ag were mixed at the given composition and melted in an arc furnace under Ar atmosphere. The melted mixtures were ground and compressed at 3 GPa and 800 °C in an h-BN container. The photoemission experiment was performed at the twin-helical undulator beamline BL25SU of SPring-8, which was equipped with a SCIENTA SES 200 analyzer.²⁴ The samples were fractured and measured in vacuum better than 3×10^{-10} Torr at 20 K. Electronic band structure calculations of Ba_8Si_{46} and $Ba_8Ag_6Si_{40}$ were performed using the full-potential linearized augmented-plane-wave (FLAPW) method^{25–27} based on the density-functional theory with the LDA.

Figure 2(a) shows the valence band spectra of $Ba_8Ag_xSi_{46-x}$ (x=0,1,3,6) measured with $h\nu$ =780 eV. The off-resonant valence band spectra are predominantly contributed by the Si states owing to the large proportion of Si in $Ba_8Ag_xSi_{46-x}$. The spectral structures at ~2-5, 6-8, and 8-12 eV in the spectrum of Ba₈Si₄₆ correspond to the Si 3*p*, 3sp, and 3s states, respectively, which are clearly observed in this spectrum, and a sharp peak is observed at E_F . By Ag doping, the Ag 4d peak arises at ~ 6.5 eV in the valence band and increases the intensity in keeping with Ag content. In addition, the peak intensity at E_F decreases with increasing Ag content and additional states grow at ~ 1.5 eV, that can be considered bonding states with Ag. In Fig. 2(b), the valence band spectra are measured in the energy range up to 2.5 eV to see the electronic states near E_F in more detail. The on-resonant photoemission spectra ($h\nu$ =789 eV) at the Ba 3d threshold were also shown in Fig. 2(b) by closed circles to study the contribution of Ba states to the valence band. In the off-resonant spectra shown by open circles in Fig. 2(b), the sharp peak at E_F is clearly seen in Ba₈Si₄₆ and drastically decreases the intensity with increasing Ag content. The additional structure is also clear in Fig. 2(b). The structure at $\sim 1.5 \text{ eV}$ which is not observed in Ba₈Si₄₆ appears in Ba₈Ag₁Si₄₅ and increases the intensity in keeping with Ag content.

To estimate the $N(E_F)$, area under the curve up to 0.35 eV is evaluated from the off-resonant spectra of Ba₈Ag_xSi_{46-x} (x=0,1,3,6). The areas normalized by the value for Ba₈Si₄₆ are estimated at 1, 0.573, 0.303, and 0.268 for x=0, 1, 3, and 6, respectively, which are plotted in Fig. 2(c) by open circles. Since $N(E_F)$ for Ba₈Si₄₆ was reported to be 31 states per eV per Si₄₆ by the specific-heat measurement⁵ $N(E_F)$ for x=1, 3, and 6 are estimated at 17.8, 9.4, and 8.3 states per eV per Si₄₆, respectively. Compared to $N(E_F)$ in alkali-metal doped C₆₀ materials which have been reported as $N(E_F)=8.5$ and 10 per eV per C₆₀ for K₃C₆₀ and Rb₃C₆₀ in Ref. 28, respectively, the values for Ba₈Ag_xSi_{46-x} are quite large in $x \le 1$ and equivalent in $x \ge 3$. It shows that the large DOS at E_F is characteristic of Ba₈Ag_xSi_{46-x} with small x and plays the important role in the appearance of the superconductivity. The BCS theory states that T_c shows the exponential decrease against $1/\lambda$.¹⁸ Hence, the reduction of $N(E_F)$ in keeping with the T_c is qualitatively consistent with the BCS theory. In Ba₈Ag_xSi_{46-x}, the reduction of $N(E_F)$ is the dominant factor causing the decrease of the T_c with the increase of Ag content. However, although the finite DOS still exists at E_F [$N(E_F) \ne 0$], no superconductivity is observed in Ba₈Ag₆Si₄₀, which may indicate that the vibrational coupling between Ba and cages consisting of Si and Ag weakens in x=6.

In the on-resonant spectra of Ba_8Si_{46} and $Ba_8Ag_1Si_{45}$, the peak at E_F is markedly enhanced. In particular, the enhancement in the on-resonant spectrum of Ba₈Si₄₆ is restricted within the narrow energy range up to ~ 0.8 eV, which is also found by the sharp pDOS of Ba at E_F in the difference spectrum [Fig. 2(d)] between on- and off-resonant spectra. It shows that the delocalized 6s state that is occupied by two electrons in the configuration of atomic Ba is not occupied in Ba_8Si_{46} . Part of these two electrons occupy the Ba 5d state to form the narrow hybridized peak at E_F together with Si 3p, and the residual electrons are donated to the Si sites in Ba_8Si_{46} . In Fig. 2(c), areas under the curves up to 0.35 eV for the on-resonant spectra are also plotted by closed circles, which show the enhancement of 65.5% in the on-resonance over the off-resonance for Ba₈Si₄₆. Ba-doped fullerene Ba_6C_{60} , which exhibits the superconductivity below 7 K,²⁹ also shows the peak near E_F originated by the hybridization between Ba 5d and t_{1u} and t_{1g} orbitals of C₆₀.^{30,31} The resonant photoemission for Ba_6C_{60} has shown the ~25% resonant enhancement of Ba states the peak near E_F .³⁰ Therefore, the proportion of the Ba 5d in the peak at E_F is much larger in Ba_8Si_{46} than in Ba_6C_{60} . This comparison verifies that the strong hybridization with Ba 5d causes the large DOS of $Ba_8Ag_xSi_{46-x}$ (x=0). As found by Fig. 2(c), the enhancement near E_F in the on-resonant spectrum of Ba₈Ag₁Si₄₅ over the off-resonant spectrum is larger than that of Ba₈Si₄₆. The difference spectra [Fig. 2(d)] also show the larger intensity near E_F in Ba₈Ag₁Si₄₅ than in Ba₈Si₄₆. Furthermore, the enhancement in the on-resonant spectrum for Ba₈Ag₁Si₄₅ spreads in wider energy range [Fig. 2(b)]. It indicates that, in Ba₈Ag₁Si₄₅, delocalized Ba 6s states are partially occupied in addition to the Ba 5d state. This additional occupation into the Ba 6s also contributes to the increase of the Ba pDOS (the intensity of the difference spectrum) near E_F .³² The offresonant spectrum of Ba₈Ag₁Si₄₅, on the other hand, shows the much lower intensity near E_F than that of Ba₈Si₄₆, which indicates the decrease of Si pDOS near E_F in Ba₈Ag₁Si₄₅. The decrease of the Si pDOS in Ba₈Ag₁Si₄₅ can be understood by the polar character of the Ag-Si bond,³³ which will be also discussed in the Si 2p core-level photoemission. In Ag-doped Si clathrates with $x \ge 3$, strong resonance is not observed near E_F but weak enhancement in the on-resonant spectra still remains in the wide energy range including the additional states at ~1.5 eV and the reduced peak at E_F [Fig. 2(b)]. The difference spectra of $x \ge 3$ in Fig. 2(d) also show that there exists the low intensity in the energy range be-

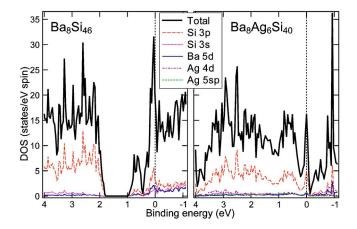


FIG. 3. (Color online.) Total and partial DOS of Ba_8Si_{46} and $Ba_8Ag_6Si_{40}$ by the LDA calculation.

tween E_F and 2.5 eV. It indicates that the Ba 5*d* electron, which forms the large Ba *p*DOS at E_F and derives the hybridization forming the large DOS at E_F in $x \le 1$, is attracted to the Ag sites, while the delocalized Ba 6*s* is left in the valence band. As the result that the hybridization with Ba 5*d* state weakens, the DOS near E_F decreases further with the increase of Ag content in $x \ge 3$.

In Figs. 3(a) and 3(b), the (*p*)DOS by LDA calculation are shown for Ba₈Si₄₆ and Ba₈Ag₆Si₄₀, respectively. The LDA calculation reproduces the drastic change of the observed valence band from Ba₈Si₄₆ to Ba₈Ag₆Si₄₀. For Ba₈Si₄₆, the sharp peak at E_F and little DOS between 1 and 2 eV shown in Fig. 2 are consistent with the calculation in Fig. 3(a). The calculation also reproduces the observed spectrum of Ba₈Ag₆Si₄₀, that is, the decrease of the peak at E_F and additional states between 1 and 2 eV. In the calculation for Ba₈Ag₆Si₄₀, the sharp DOS which show the large contribution from Ba 5*d* is expected to be located above E_F (~-1 eV). Hence, a part of the DOS at the E_F in Ba₈Si₄₆ moves above E_F in Ba₈Ag₆Si₄₀ and the other forms the bonding states located at the energy range between the peak at E_F and Si 3*p* state.

Figure 4(a) shows the Ba 4d core-level spectra of $Ba_8Ag_xSi_{46-x}$ (x=0,1,3,6). The Ba 4d spectra reveal the two sets of spin-orbit splitting peaks which is labeled as A and B in Fig. 4(a). It is found by the intensity ratio between A and B that the main peaks A at 89.7 eV and 92.4 eV are originated by Ba inside the Si_{24} cage (six sites in Si_{46} unit) and smaller peaks B at 89.2 and 91.8 eV come from Ba inside the Si₂₀ cage (two sites in Si_{46} unit). Hence, the valence of Ba is not homogeneous between Si_{24} and Si_{20} cages, but depends on the number of surrounding Si atoms. In the Ba₆C₆₀ mentioned above, Ba is thought to be monovalent,³¹ which indicates that the valence of Ba deviates from the limit of complete (two electrons) charge transfer from 6s because t_{1u} and t_{1g} orbitals of C₆₀ are hybridized with Ba 5d state. The peak position of Ba $4d_{5/2}$ in Ba₆C₆₀ has been ~89.8 eV,³⁰ which is approximately equal to the peak position A (89.725 eV) of Ba $4d_{5/2}$ in Ba₈Si₄₆. Therefore, Ba inside the Si₂₄ cage in Ba_8Si_{46} possibly shows the configuration similar to Ba in Ba₆C₆₀. The low-energy component B originated by Ba inside the Si₂₀ cage shows that the valence (the number of

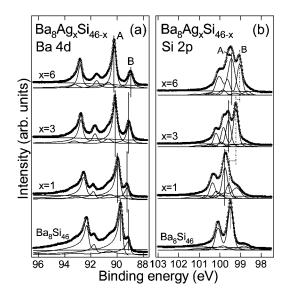


FIG. 4. (a) The open circles show the Ba 4*d* spectra of $Ba_8Ag_xSi_{46-x}$ (*x*=0,1,3,6), measured with $h\nu$ =815 eV. The fitting results are shown by the solid lines, which consist of three components. The peaks marked with A and B in the figure are derived from Ba inside the Si₂₄ and Si₂₀ cages, respectively. The broad component corresponds to the contribution of the surface. (b) The open circles show the Si 2*p* spectra of $Ba_8Ag_xSi_{46-x}$ (*x*=0,1,3,6) measured with $h\nu$ =815 eV. The fitting curves are shown by the solid lines. In the spectra of Ag-doped Si clathrates, the main components are marked by A and B.

occupied 5*d* electrons) of Ba inside the Si_{20} cage is more than that of Ba inside the Si₂₄ cage for Ba₈Si₄₆ in which only the Ba 5d state is partially occupied. Hence, the Ba 5d electrons used for the hybridization at the Si₂₀ cage are more than that at the Si₂₄ cage. This excess electron at Ba inside the Si₂₀ cage causes the large enhancement of the onresonant valence band spectrum in Ba₈Si₄₆ compared to that in Ba₆C₆₀. With increasing Ag content, peak A shifts to higher binding energy and peak B shifts to lower binding energy, while the peak B in Ba₈Ag₁Si₄₅ slightly shifts to higher binding energy than that in Ba₈Si₄₆ in contrast to the shift of peak B in $x \ge 3$. The high binding energy shifts of both the peaks in Ba₈Ag₁Si₄₅ may indicate that the less occupation of the Ba electrons at the Si_{24} and Si_{20} cages in $Ba_8Ag_1Si_{45}$ than in Ba_8Si_{46} . The on-resonant spectrum of Ba₈Ag₁Si₄₅, however, shows that the delocalized Ba 6s state is added in the valence band. This behavior in the valence band is consistently explained by the characteristic feature of Ba core level. The Ba 4d core level is probably more sensitive to the variation in the number of 5d electrons than that of the 6s electrons as has been interpreted in Refs. 34-36, since the Ba 5d orbitals have smaller radial extent than the delocalized 6s orbital. Hence, the shifts of the peaks A and B in the Ba 4d spectrum of $Ba_8Ag_1Si_{45}$ are governed by the decrease of the 5d electron and insensitive to the change in the occupation of 6s. The high binding energy shifts for the peaks A and B of the Ba 4d spectrum and the broad enhancement of the on-resonant spectrum in Ba₈Ag₁Si₄₅, therefore, indicate that Ba in $Ba_8Ag_1Si_{45}$ loses the 5d electrons in the Si₂₄ and Si₂₀ cages and gains the 6s character in the valence

band. As the result that the hybridization between Ba 5*d* and Si 3*p* weakens, Ba in $Ba_8Ag_1Si_{45}$ may show a configuration similar to atomic Ba.

In clathrates with $x \ge 3$, the observed core-level shift of peak A indicates that Ba inside the Si_{24} cages loses the 5d electrons further with increasing Ag content. The decrease of the Ba 5d electron at the Si_{24} cage can be understood by considering the interchange of the charge with doped Ag. Since Ag has greater electronegativity than Ba and Si,³³ the 5*d* electron of Ba inside the Si_{24} cage including doped Ag is attracted to the Ag site further. Therefore, the 5d electron of Ba inside the Si₂₄ cage decreases selectively. The decrease of the Ba 5d electrons in Ag-doped clathrates with $x \ge 3$ is consistent with the resonant photoemission result, showing the reduction of the Ba 5d states at E_F with the increase of Ag content in $x \ge 3$. The shift of peak B is opposite to that of peak A in the Ba 4*d* spectra of $x \ge 3$. The difference between the valences of Ba in the Si₂₄ and Si₂₀ cages, therefore, becomes large with increasing Ag content. The shift of peak B in $x \ge 3$ indicates that the electrons that move to the Si network from Ba inside the Si₂₄ cage through doped Ag are used for the hybridization between Si 3p and Ba 5d states at the Si₂₀ cage in clathrates of $x \ge 3$. Nevertheless, the total number of Ba 5d electrons decreases with the increase of Ag content in $x \ge 3$ as observed in the resonant photoemission. Since the number of Ba atoms inside the Si₂₄ cage (six sites in Si₄₆ unit) is three times as large as that inside the Si₂₀ cage (two sites in Si_{46} unit), the change in the valence of Ba inside the Si_{24} cage affects the valence band more strongly than that of Ba inside the Si₂₀ cage. Furthermore, the decrease in the Ba 5*d* electron at the Si_{24} cage is larger than the increase in that at the Si₂₀ cage, because the positive shift of peak A, which is +0.500 eV in Ba₈Ag₆Si₄₀ from the peak position in Ba₈Si₄₆, is larger than the negative shift of peak B which is -0.175 eV.

Another point to note is that the Ba 4d peak of Ba_8Si_{46} takes on an asymmetric shape, which can be distinguished by the tail extended up to the background, and becomes symmetric with doping Ag. An asymmetric line shape results from sharp cut off at E_F and is described by Doniach and Šunjić line shape.³⁷ To estimate the asymmetric parameter, a curve fitting was performed using the Doniach and Šunjić line shape.³⁷ The best fitting results are shown in Fig. 4(a) by solid curves. Each fitting result consists of three spin-orbit split curves; the two asymmetric curves correspond to the peaks due to Ba in the Si₂₄ and Si₂₀ cages, as already shown by A and B, respectively, and the other is the broad peak with smaller amount, which can be considered a surface component. The conclusive fitting curves match well with the photoemission spectra. The asymmetric parameters of peak A are estimated to be 0.375, 0.250, 0.130, and 0.080 for *x*=0, 1, 3, and 6 by the curve fitting, respectively. The systematic transformation of the Ba 4d spectra from asymmetric shape for Ba₈Si₄₆ into more symmetric shape for Ba₈Ag₆Si₄₀ is consistent with the changes in the DOS at E_F and the T_c depending on Ag content.

Figure 4(b) shows the Si 2p spectra of Ba₈Ag_xSi_{46-x} (x=0,1,3,6). Although the Si 2p spectra are complicated in Ag-doped clathrates, a curve fitting was attempted to understand the spectral features. In Ba₈Si₄₆, there are three unequal

sites, which are namely 24k, 16i, and 6c sites. The 6c site is the site which is represented by black circles in Fig. 1 and the 24k site is adjacent to the 6c site. The second nearest neighbor to the 6c site is the 16i site. In spite of the presence of these different Si sites, the Si 2p spectrum of Ba₈Si₄₆ shows one bulk component with large intensity and small contributions located at the low and high binding energy sides to the main peak [Fig. 4(b)]. The only one large component in Si 2p spectrum indicates that the charge is homogeneously distributed on Si network in Ba₈Si₄₆, in contrast to the difference between the valences of Ba in the Si_{24} and Si_{20} cages. The fitting spectra for Ag-doped Si clathrates show two main components. Hence, Ag doping causes the Si atoms with two different valences. In Fig. 4(b) the main component at the high binding energy side is marked with A and that at the low binding energy side with B. In Ba₈Ag₁Si₄₅, peak A is located at higher binding energy than the main peak of Ba₈Si₄₆. The bonding between Ag and Si is expected to have the larger electron density at the Ag site because the electronegativity of Ag is still greater than that of Si.33 Hence, the Si 2p spectrum of Ba₈Ag₁Si₄₅ shows the component which shifts to the higher binding energy. The peak shift of the Si 2p spectrum in Ba₈Ag₁Si₄₅ is consistent with the valence band spectrum of Ba8Ag1Si45, which shows the drastic decrease in the pDOS of Si. In $x \ge 3$, on the other hand, the component (B), which greatly shifts to low binding energy, is observed in the Si 2p spectra. Since Ba provides the electrons for the Si network at the Si₂₄ cage through doped Ag in $x \ge 3$ as was shown in the peak shift (A) of the Ba 4d spectra, the excess electron charge on Ag possibly causes conversion of the bonding character so as to provide the electrons for the Si network through Ag in $x \ge 3$. The provided electrons to Si sites are partially used for the hybridization with Ba at the Si₂₀ cage as was shown by the shift of peak B in Ba 4d spectra and are probably also added to the bonding states at ~ 1.5 eV in the valence band. Hence, complicated charge balance depending on Ag content results so as to stabilize $Ba_8Ag_xSi_{46-x}$ energetically. In $Ba_8Ag_6Si_{40}$, since all of the 6c sites are replaced with Ag atoms, the Si 2pspectrum is simply assigned by the Si sites. If components A and B are assumed to come from the 24k and 16i sites, the ratio between A and B is expected to be 0.6:0.4, which approximately matches with the intensity ratio 0.52:0.48 estimated by the curve fit in $Ba_8Ag_6Si_{40}$.

In conclusion, valence band and core-level photoemission using soft x rays have been applied to $Ba_8Ag_xSi_{46-x}$ (x=0,1,3,6). In the valence band spectra of Ba₈Si₄₆, a sharp peak was prominently observed at E_F . The resonant photoemission confirmed the strong hybridization with the Ba 5dstate for this peak. The peak intensity at E_F decreases with increasing Ag content. In Ba₈Ag₁Si₄₅, the valence electron of Si is attracted to Ag, which causes the decrease in the DOS at E_F . The resonant photoemission and Ba 4d core-level spectrum have shown that the Ba 6s state admixes in the valence band of Ba₈Ag₁Si₄₅. In $x \ge 3$, conversion of the bonding character takes place. In contrast to Ba₈Ag₁Si₄₅, the 5d electron of Ba inside the Si₂₄ cage is exclusively attracted to Ag in $x \ge 3$ and the DOS decreases further as the result that the hybridization with the Ba 5d state weakens. The excess electron charge on Ag flows to Si network. The electron-flow to Si network would form the additional bonding states at \sim 1.5 eV with doped Ag. As the result of these interchanges of the charge between the elements of $Ba_8Ag_xSi_{46-x}$, Ag doping causes the decrease in the peak intensity at E_F , which is the dominant factor decreasing the T_c . The absence of superconductivity in Ba₈Ag₆Si₄₀ for which $N(E_F) \neq 0$ indicates the weak vibrational coupling between Ba and cages consisting of Si and Ag in Ba₈Ag₆Si₄₀. The change of the valence band by Ag doping was consistent with the LDA calculation. $Ba_{8-x}Si_{46}$ (Refs. 6 and 7) and $Na_xBa_{8-x}Si_{46}$ (Refs. 2 and 3) in which Ba atoms are deintercalated and partially replaced by Na atoms, respectively, and $Ba_8Si_{46-x}Ge_x$ (Ref. 38) in which Si atoms are partially replaced with Ge atoms also show the variation of T_c . Therefore, the experimental studies of the electronic states for these systems are desirable for understanding the overall mechanism on the appearance of the narrow peak at E_F and superconductivity in Si clathrate compounds.

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