## **Role of Ag doping in Ba<sub>8</sub>Si<sub>46</sub> compounds**

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The silicon clathrate compound  $Ba_8Si_{46}$  shows superconductivity below the critical temperature  $(T_c)$  of 8 K, and the  $T_c$  decreases monotonically with doping Ag. In order to reveal effects of Ag doping on the electronic states, we have applied soft x-ray photoemission spectroscopy to Ag-doped silicon clathrate compounds  $Ba_8Ag_xSi_{46-x}$  ( $x=0,1,3,6$ ). The valence band photoemission spectra show that a Ba 5*d*-derived state at the Fermi level  $(E_F)$ , which is prominently observed in  $Ba_8Si_{46}$ , decreases with increasing Ag content. The reduction in the peak intensity at  $E_F$  with increasing Ag content is therefore in accord with the decrease of  $T_c$  in Ba8Ag*x*Si46−*x*. Band structure calculation using local-density approximation reproduces the observed valence band spectra of  $x=0$  and 6. The Si 2*p* and Ba 4*d* core-level photoemission spectra demonstrate that the valence electron of Si is attracted to the Ag site in  $x=1$  and the 5*d* electron of Ba inside the Si<sub>24</sub> cage is further donated to Ag in  $x \ge 3$ . Hence, Ag doping leads to the reduction of the peak at  $E_F$ .

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Studies on silicon clathrate compounds trace back to discussion on the metal-insulator transition in  $Na<sub>x</sub>Si<sub>46</sub>$  and  $Na_xSi1_{36}$ <sup>1</sup> Recently,  $Na_2Ba_6Si_{46}$  and  $Ba_8Si_{46}$  have been found to show superconductivity with  $T_c$  of 4 and 8 K, respectively, $2^{-9}$  which have led to much notice into Si clathrate compounds. In  $Ba_8Si_{46}$ , Si atoms form a network consisting of  $Si<sub>24</sub>$  and  $Si<sub>20</sub>$  cages and Ba atoms are located at the centers of these Si cages (Fig. 1).  $Ba_8Si_{46}$  is similar to alkalimetal doped  $C_{60}$  materials in terms of a polyhedra-network system showing superconductivity.<sup>10–15</sup> In the C<sub>60</sub> materials, the polyhedra are held together by van der Waals–type interactions. The superconductivity in the alkali-metal-doped  $C_{60}$ materials is based on the competition between the on-site Coulomb repulsion energy among the electrons occupying  $t_{1u}$ -derived  $C_{60}$  bands and the coupling between these electrons and intramolecular vibrations of  $C_{60}$ . In Si clathrate compounds, however, the polyhedra share the pentagonal and hexagonal faces as shown in Fig. 1 and form a covalently bonded network. The band structure calculation by the density functional theory for  $Ba_8Si_{46}$  has predicted that strong hybridization between the Si 3*p* and Ba 5*d* states modifies the density of states (DOS) of the Si network and forms a narrow peak at the  $E_F$ .<sup>7,16,17</sup> This hybridized state at  $E_F$  is thought to play a crucial role in the superconductivity of Ba<sub>8</sub>Si<sub>46</sub>, since  $T_c$  is a function of DOS at  $E_F$  in normal state  $[N(E_F)]$  through the coupling constant  $\lambda = N(E_F)V$ , where  $V$  is electron-excitation coupling strength.<sup>18</sup> The photoemission spectroscopy using He I $\alpha$  (21.218 eV) and He  $\Pi\alpha$  (40.814 eV) has shown the narrow peak at  $E_F$  and has observed the superconducting gap  $[2\Delta(E_F)/k_B T_c]$  of 4.38 in the superconducting state.<sup>19</sup> The hybridized state at  $E_F$  has been also observed by the Knight shift in nuclear magnetic resonance  $(NMR)^{20,21}$  and the specific-heat and magnetic susceptibility measurements.<sup>5</sup> On the other hand, the study

by neutron scattering and extended x-ray absorption fine structure (EXAFS) has shown that the vibrational coupling between Ba and Si network inside the  $Si<sub>24</sub>$  cage is also the important parameter to explain consistently the superconductivity in  $Ba_8Si_{46}$  and the absence of superconductivity in  $\text{Na}_8\text{Si}_{46}$  for which  $N(E_F) \neq 0.22$  The recent study on the isotope effect in Raman spectrum has indicated that the superconductivity in  $Ba_8Si_{46}$  is explained by Bardeen-Cooper-Schrieffer (BCS) mechanism with medium coupling regime.<sup>5</sup>

In Ba<sub>8</sub>Ag<sub>x</sub>Si<sub>46−*x*</sub> studied here, Si atoms are replaced with Ag atoms at  $6c$  sites which belong only to the  $Si<sub>24</sub>$  cages. The 6*c* sites are represented by black circles in Fig. 1 which shows the crystal structure of  $Ba_8Ag_xSi_{46-x}$ . In  $Ba_8Ag_3Si_{43}$ , Ag atoms occupy the 6*c* sites alternately in the Si sixfold ring. Ba<sub>8</sub>Ag<sub>x</sub>Si<sub>46−*x*</sub> shows the superconductivity in  $x \le 3$ , while the  $T_c$  decreases with increasing Ag content to 7 K and 2.8 K for  $x=1$  and 3, respectively. The superconductivity has not been observed down to 1.8 K in Ag-doped clathrates with  $x > 3$ .<sup>23</sup> Therefore, the systematic study for the elec-



FIG. 1. Crystal structure of Ba8Ag*x*Si46−*x*. Black circles indicate the 6*c* sites where Si atom is replaced with Ag atom by Ag doping.



FIG. 2. (a) The valence band spectra of  $Ba_8Ag_xSi_{46-x}$  ( $x=0,1,3,6$ ), measured with  $h\nu=780$  eV. The spectra are normalized by the intensity at 12.0 eV. (b) Resonant photoemission spectra at Ba  $3d$  threshold. The closed and open circles show the on-resonant  $(h\nu)$ =789 eV) and off-resonant ( $h\nu$ =780 eV) spectra of Ba<sub>8</sub>Ag<sub>x</sub>Si<sub>46−*x*</sub> (*x*=0,1,3,6). The spectra are normalized by the photon flux. (c) The areas under the curves between  $E_F$  and 0.35 eV of the on- and off-resonant spectra shown in (b) are plotted by the closed and open circles with error bars, respectively. All values are normalized by the value for the off-resonant spectrum of  $Ba_8Si_{46}$ . (d) The difference spectra between the on- and off-resonance spectra of  $Ba_8Ag_xSi_{46-x}$  ( $x=0,1,3,6$ ).

tronic states of Ba<sub>8</sub>Ag<sub>*x*</sub>Si<sub>46−*x*</sub> as a function of Ag content  $(x)$ provides the important knowledge on the superconductivity in Si clathrate compounds such as the mechanism of the change in the  $T_c$  and the appearance and absence of the superconductivity, which are indeed related to the behavior of the DOS at  $E_F$ . Here, in order to study the electronic states of  $Ba_8Ag_xSi_{46-x}$  depending on Ag content  $(x)$  and to elucidate the change in them caused by Ag doping, we have applied soft x-ray photoemission to  $Ba_8Ag_xSi_{46-x}$  with  $x=0, 1, 3$ , and 6. An advantage in the use of soft x-ray is a long photoelectron mean free path compared to the photoemission using ultraviolet rays and, hence, the effective probe of bulk electronic states. The valence band photoemission has probed the DOS at  $E_F$  depending on Ag content and has shown the additional bonding states at  $\sim$ 1.5 eV binding energy in Agdoped Si clathrates. In addition, the resonant photoemission at Ba 3*d* threshold, which enhances the Ba states selectively, has been performed to study the partial density of states  $(pDOS)$  of Ba in the valence band. Ba  $4d$  and Si  $2p$  corelevel photoemission have been also done to study the electronic states of  $Ba_8Ag_xSi_{46-x}$  ( $x=0,1,3,6$ ) with element specification. These core-level spectra elucidate the interchange of the charge between the elements and the origin of changes in the valence band.

 $Ba_8Ag_xSi_{46-x}$   $(x=0,1,3,6)$  were prepared under high pressure.<sup>4,6,7,23</sup> Ba, Si, and Ag were mixed at the given composition and melted in an arc furnace under Ar atmosphere. The melted mixtures were ground and compressed at 3 GPa and 800 °C in an h-BN container. The photoemission experiment was performed at the twin-helical undulator beamline BL25SU of SPring-8, which was equipped with a SCIENTA SES 200 analyzer.<sup>24</sup> The samples were fractured and measured in vacuum better than  $3 \times 10^{-10}$  Torr at 20 K. Electronic band structure calculations of  $Ba_8Si_{46}$  and  $Ba_8Ag_6Si_{40}$ were performed using the full-potential linearized augmented-plane-wave (FLAPW) method $25-27$  based on the density-functional theory with the LDA.

Figure  $2(a)$  shows the valence band spectra of  $Ba_8Ag_xSi_{46-x}$  ( $x=0,1,3,6$ ) measured with  $h\nu=780$  eV. The off-resonant valence band spectra are predominantly contributed by the Si states owing to the large proportion of Si in Ba<sub>8</sub>Ag<sub>*x*</sub>Si<sub>46−*x*</sub>. The spectral structures at  $\sim$ 2−5, 6–8, and  $8-12$  eV in the spectrum of Ba<sub>8</sub>Si<sub>46</sub> correspond to the Si 3*p*, 3*sp*, and 3*s* states, respectively, which are clearly observed in this spectrum, and a sharp peak is observed at  $E_F$ . By Ag doping, the Ag 4*d* peak arises at  $\sim 6.5$  eV in the valence band and increases the intensity in keeping with Ag content. In addition, the peak intensity at  $E_F$  decreases with increasing Ag content and additional states grow at  $\sim$ 1.5 eV, that can be considered bonding states with Ag. In Fig. 2(b), the valence band spectra are measured in the energy range up to 2.5 eV to see the electronic states near  $E_F$  in more detail. The on-resonant photoemission spectra ( $h\nu$ =789 eV) at the Ba  $3d$  threshold were also shown in Fig. 2(b) by closed circles to study the contribution of Ba states to the valence band. In the off-resonant spectra shown by open circles in Fig. 2(b), the sharp peak at  $E_F$  is clearly seen in  $Ba_8Si_{46}$  and drastically decreases the intensity with increasing Ag content. The additional structure is also clear in Fig.  $2(b)$ . The structure at  $\sim$ 1.5 eV which is not observed in Ba<sub>8</sub>Si<sub>46</sub> appears in  $Ba_8Ag_1Si_{45}$  and increases the intensity in keeping with Ag content.

To estimate the  $N(E_F)$ , area under the curve up to 0.35 eV is evaluated from the off-resonant spectra of  $Ba_8Ag_xSi_{46-x}$  $(x=0, 1, 3, 6)$ . The areas normalized by the value for  $Ba_8Si_{46}$ are estimated at 1, 0.573, 0.303, and 0.268 for *x*=0, 1, 3, and  $6$ , respectively, which are plotted in Fig.  $2(c)$  by open circles. Since  $N(E_F)$  for Ba<sub>8</sub>Si<sub>46</sub> was reported to be 31 states per eV per Si<sub>46</sub> by the specific-heat measurement<sup>5</sup>  $N(E_F)$  for  $x=1, 3$ , and 6 are estimated at 17.8, 9.4, and 8.3 states per eV per  $Si<sub>46</sub>$ , respectively. Compared to  $N(E<sub>F</sub>)$  in alkali-metal doped  $C_{60}$  materials which have been reported as  $N(E_F) = 8.5$  and 10 per eV per  $C_{60}$  for  $K_3C_{60}$  and  $Rb_3C_{60}$  in Ref. 28, respec-

tively, the values for Ba<sub>8</sub>Ag<sub>*x*</sub>Si<sub>46−*x*</sub> are quite large in  $x \le 1$ and equivalent in  $x \geq 3$ . It shows that the large DOS at  $E_F$  is characteristic of  $Ba_8Ag_xSi_{46-x}$  with small *x* and plays the important role in the appearance of the superconductivity. The BCS theory states that  $T_c$  shows the exponential decrease against  $1/\lambda$ .<sup>18</sup> Hence, the reduction of  $N(E_F)$  in keeping with the  $T_c$  is qualitatively consistent with the BCS theory. In Ba<sub>8</sub>Ag<sub>*x*</sub>Si<sub>46−*x*</sub>, the reduction of  $N(E_F)$  is the dominant factor causing the decrease of the  $T_c$  with the increase of Ag content. However, although the finite DOS still exists at  $E_F$  [ $N(E_F) \neq 0$ ], no superconductivity is observed in  $Ba_8Ag_6Si_{40}$ , which may indicate that the vibrational coupling between Ba and cages consisting of Si and Ag weakens in *x*=6.

In the on-resonant spectra of  $Ba_8Si_{46}$  and  $Ba_8Ag_1Si_{45}$ , the peak at  $E_F$  is markedly enhanced. In particular, the enhancement in the on-resonant spectrum of  $Ba_8Si_{46}$  is restricted within the narrow energy range up to  $\sim 0.8$  eV, which is also found by the sharp  $p$ DOS of Ba at  $E_F$  in the difference spectrum [Fig. 2(d)] between on- and off-resonant spectra. It shows that the delocalized 6*s* state that is occupied by two electrons in the configuration of atomic Ba is not occupied in Ba8Si46. Part of these two electrons occupy the Ba 5*d* state to form the narrow hybridized peak at  $E_F$  together with Si  $3p$ , and the residual electrons are donated to the Si sites in  $Ba_8Si_{46}$ . In Fig. 2(c), areas under the curves up to 0.35 eV for the on-resonant spectra are also plotted by closed circles, which show the enhancement of 65.5% in the on-resonance over the off-resonance for  $Ba_8Si_{46}$ . Ba-doped fullerene  $Ba_6C_{60}$ , which exhibits the superconductivity below 7 K,<sup>29</sup> also shows the peak near  $E_F$  originated by the hybridization between Ba 5*d* and  $t_{1u}$  and  $t_{1g}$  orbitals of  $C_{60}$ .<sup>30,31</sup> The resonant photoemission for Ba<sub>6</sub>C<sub>60</sub> has shown the  $\sim$ 25% resonant enhancement of Ba states the peak near  $E_F$ .<sup>30</sup> Therefore, the proportion of the Ba  $5d$  in the peak at  $E_F$  is much larger in Ba<sub>8</sub>Si<sub>46</sub> than in Ba<sub>6</sub>C<sub>60</sub>. This comparison verifies that the strong hybridization with Ba 5*d* causes the large DOS of  $Ba_8Ag_xSi_{46-x}$  ( $x=0$ ). As found by Fig. 2(c), the enhancement near  $E_F$  in the on-resonant spectrum of  $Ba_8Ag_1Si_{45}$  over the off-resonant spectrum is larger than that of  $Ba_8Si_{46}$ . The difference spectra  $[Fig. 2(d)]$  also show the larger intensity near  $E_F$  in Ba<sub>8</sub>Ag<sub>1</sub>Si<sub>45</sub> than in Ba<sub>8</sub>Si<sub>46</sub>. Furthermore, the enhancement in the on-resonant spectrum for  $Ba_8Ag_1Si_{45}$  spreads in wider energy range  $[Fig. 2(b)].$  It indicates that, in Ba8Ag1Si45, delocalized Ba 6*s* states are partially occupied in addition to the Ba 5*d* state. This additional occupation into the Ba 6*s* also contributes to the increase of the Ba *p*DOS (the intensity of the difference spectrum) near  $E_F$ .<sup>32</sup> The offresonant spectrum of  $Ba_8Ag_1Si_{45}$ , on the other hand, shows the much lower intensity near  $E_F$  than that of  $Ba_8Si_{46}$ , which indicates the decrease of Si  $p$ DOS near  $E_F$  in Ba<sub>8</sub>Ag<sub>1</sub>Si<sub>45</sub>. The decrease of the Si  $pDOS$  in  $Ba_8Ag_1Si_{45}$  can be understood by the polar character of the Ag-Si bond,  $33$  which will be also discussed in the Si 2*p* core-level photoemission. In Ag-doped Si clathrates with  $x \ge 3$ , strong resonance is not observed near  $E_F$  but weak enhancement in the on-resonant spectra still remains in the wide energy range including the additional states at  $\sim$  1.5 eV and the reduced peak at  $E_F$  [Fig. 2(b)]. The difference spectra of  $x \ge 3$  in Fig. 2(d) also show that there exists the low intensity in the energy range be-



FIG. 3. (Color online.) Total and partial DOS of Ba<sub>8</sub>Si<sub>46</sub> and  $Ba_8Ag_6Si_{40}$  by the LDA calculation.

tween  $E_F$  and 2.5 eV. It indicates that the Ba 5*d* electron, which forms the large Ba  $p$ DOS at  $E_F$  and derives the hybridization forming the large DOS at  $E_F$  in  $x \le 1$ , is attracted to the Ag sites, while the delocalized Ba 6*s* is left in the valence band. As the result that the hybridization with Ba 5*d* state weakens, the DOS near  $E_F$  decreases further with the increase of Ag content in  $x \ge 3$ .

In Figs.  $3(a)$  and  $3(b)$ , the  $(p)$ DOS by LDA calculation are shown for  $Ba_8Si_{46}$  and  $Ba_8Ag_6Si_{40}$ , respectively. The LDA calculation reproduces the drastic change of the observed valence band from  $Ba_8Si_{46}$  to  $Ba_8Ag_6Si_{40}$ . For  $Ba_8Si_{46}$ , the sharp peak at  $E_F$  and little DOS between 1 and 2 eV shown in Fig. 2 are consistent with the calculation in Fig.  $3(a)$ . The calculation also reproduces the observed spectrum of  $Ba_8Ag_6Si_{40}$ , that is, the decrease of the peak at  $E_F$  and additional states between 1 and 2 eV. In the calculation for  $Ba_8Ag_6Si_{40}$ , the sharp DOS which show the large contribution from Ba 5*d* is expected to be located above  $E_F$  $(\sim -1$  eV). Hence, a part of the DOS at the  $E_F$  in Ba<sub>8</sub>Si<sub>46</sub> moves above  $E_F$  in Ba<sub>8</sub>Ag<sub>6</sub>Si<sub>40</sub> and the other forms the bonding states located at the energy range between the peak at  $E_F$ and Si 3*p* state.

Figure  $4(a)$  shows the Ba  $4d$  core-level spectra of  $Ba_8Ag_xSi_{46-x}$  ( $x=0,1,3,6$ ). The Ba 4*d* spectra reveal the two sets of spin-orbit splitting peaks which is labeled as A and B in Fig. 4(a). It is found by the intensity ratio beween A and B that the main peaks A at 89.7 eV and 92.4 eV are originated by Ba inside the  $Si<sub>24</sub>$  cage (six sites in  $Si<sub>46</sub>$  unit) and smaller peaks B at 89.2 and 91.8 eV come from Ba inside the  $Si<sub>20</sub>$ cage (two sites in  $Si_{46}$  unit). Hence, the valence of Ba is not homogeneous between  $Si<sub>24</sub>$  and  $Si<sub>20</sub>$  cages, but depends on the number of surrounding Si atoms. In the  $Ba_6C_{60}$  mentioned above, Ba is thought to be monovalent, $31$  which indicates that the valence of Ba deviates from the limit of complete (two electrons) charge transfer from 6*s* because  $t_{1u}$  and  $t_{1g}$  orbitals of C<sub>60</sub> are hybridized with Ba 5*d* state. The peak position of Ba  $4d_{5/2}$  in Ba<sub>6</sub>C<sub>60</sub> has been ~89.8 eV,<sup>30</sup> which is approximately equal to the peak position  $A(89.725 \text{ eV})$  of Ba  $4d_{5/2}$  in Ba<sub>8</sub>Si<sub>46</sub>. Therefore, Ba inside the Si<sub>24</sub> cage in  $Ba_8Si_{46}$  possibly shows the configuration similar to Ba in  $Ba<sub>6</sub>C<sub>60</sub>$ . The low-energy component B originated by Ba inside the  $Si<sub>20</sub>$  cage shows that the valence (the number of



FIG. 4. (a) The open circles show the Ba 4*d* spectra of  $Ba_8Ag_xSi_{46-x}$  (*x*=0,1,3,6), measured with  $h\nu=815$  eV. The fitting results are shown by the solid lines, which consist of three components. The peaks marked with A and B in the figure are derived from Ba inside the  $Si<sub>24</sub>$  and  $Si<sub>20</sub>$  cages, respectively. The broad component corresponds to the contribution of the surface. (b) The open circles show the Si 2*p* spectra of Ba<sub>8</sub>Ag<sub>*x*</sub>Si<sub>46−*x*</sub> (*x*=0,1,3,6) measured with  $h\nu=815$  eV. The fitting curves are shown by the solid lines. In the spectra of Ag-doped Si clathrates, the main components are marked by A and B.

occupied  $5d$  electrons) of Ba inside the  $Si<sub>20</sub>$  cage is more than that of Ba inside the  $Si<sub>24</sub>$  cage for  $Ba<sub>8</sub>Si<sub>46</sub>$  in which only the Ba 5*d* state is partially occupied. Hence, the Ba 5*d* electrons used for the hybridization at the  $Si<sub>20</sub>$  cage are more than that at the  $Si<sub>24</sub>$  cage. This excess electron at Ba inside the  $Si<sub>20</sub>$  cage causes the large enhancement of the onresonant valence band spectrum in  $Ba_8Si_{46}$  compared to that in  $Ba<sub>6</sub>C<sub>60</sub>$ . With increasing Ag content, peak A shifts to higher binding energy and peak B shifts to lower binding energy, while the peak B in  $Ba_8Ag_1Si_{45}$  slightly shifts to higher binding energy than that in  $Ba_8Si_{46}$  in contrast to the shift of peak B in  $x \ge 3$ . The high binding energy shifts of both the peaks in  $Ba_8Ag_1Si_{45}$  may indicate that the less occupation of the Ba electrons at the  $Si<sub>24</sub>$  and  $Si<sub>20</sub>$  cages in  $Ba_8Ag_1Si_{45}$  than in  $Ba_8Si_{46}$ . The on-resonant spectrum of Ba8Ag1Si45, however, shows that the delocalized Ba 6*s* state is added in the valence band. This behavior in the valence band is consistently explained by the characteristic feature of Ba core level. The Ba 4*d* core level is probably more sensitive to the variation in the number of 5*d* electrons than that of the 6*s* electrons as has been interpreted in Refs. 34–36, since the Ba 5*d* orbitals have smaller radial extent than the delocalized 6*s* orbital. Hence, the shifts of the peaks A and B in the Ba 4*d* spectrum of  $Ba_8Ag_1Si_{45}$  are governed by the decrease of the 5*d* electron and insensitive to the change in the occupation of 6*s*. The high binding energy shifts for the peaks A and B of the Ba 4*d* spectrum and the broad enhancement of the on-resonant spectrum in  $Ba_8Ag_1Si_{45}$ , therefore, indicate that Ba in  $Ba_8Ag_1Si_{45}$  loses the 5*d* electrons in the  $Si<sub>24</sub>$  and  $Si<sub>20</sub>$  cages and gains the 6*s* character in the valence

band. As the result that the hybridization between Ba 5*d* and Si  $3p$  weakens, Ba in Ba<sub>8</sub>Ag<sub>1</sub>Si<sub>45</sub> may show a configuration similar to atomic Ba.

In clathrates with  $x \ge 3$ , the observed core-level shift of peak A indicates that Ba inside the  $Si<sub>24</sub>$  cages loses the 5*d* electrons further with increasing Ag content. The decrease of the Ba  $5d$  electron at the  $Si<sub>24</sub>$  cage can be understood by considering the interchange of the charge with doped Ag. Since Ag has greater electronegativity than Ba and  $Si$ <sup>33</sup>, the  $5d$  electron of Ba inside the  $Si<sub>24</sub>$  cage including doped Ag is attracted to the Ag site further. Therefore, the 5*d* electron of Ba inside the  $Si<sub>24</sub>$  cage decreases selectively. The decrease of the Ba 5*d* electrons in Ag-doped clathrates with  $x \ge 3$  is consistent with the resonant photoemission result, showing the reduction of the Ba  $5d$  states at  $E_F$  with the increase of Ag content in  $x \ge 3$ . The shift of peak B is opposite to that of peak A in the Ba 4*d* spectra of  $x \ge 3$ . The difference between the valences of Ba in the  $Si<sub>24</sub>$  and  $Si<sub>20</sub>$  cages, therefore, becomes large with increasing Ag content. The shift of peak B in  $x \geq 3$  indicates that the electrons that move to the Si network from Ba inside the  $Si<sub>24</sub>$  cage through doped Ag are used for the hybridization between Si 3*p* and Ba 5*d* states at the Si<sub>20</sub> cage in clathrates of  $x \ge 3$ . Nevertheless, the total number of Ba 5*d* electrons decreases with the increase of Ag content in  $x \geq 3$  as observed in the resonant photoemission. Since the number of Ba atoms inside the  $Si<sub>24</sub>$  cage (six sites in  $Si<sub>46</sub>$  unit) is three times as large as that inside the  $Si<sub>20</sub>$  cage (two sites in  $Si_{46}$  unit), the change in the valence of Ba inside the  $Si<sub>24</sub>$  cage affects the valence band more strongly than that of Ba inside the  $Si<sub>20</sub>$  cage. Furthermore, the decrease in the Ba  $5d$  electron at the  $Si<sub>24</sub>$  cage is larger than the increase in that at the  $Si<sub>20</sub>$  cage, because the positive shift of peak A, which is +0.500 eV in  $Ba_8Ag_6Si_{40}$  from the peak position in  $Ba_8Si_{46}$ , is larger than the negative shift of peak B which is −0.175 eV.

Another point to note is that the Ba  $4d$  peak of  $Ba_8Si_{46}$ takes on an asymmetric shape, which can be distinguished by the tail extended up to the background, and becomes symmetric with doping Ag. An asymmetric line shape results from sharp cut off at  $E_F$  and is described by Doniach and Šunjić line shape.37 To estimate the asymmetric parameter, a curve fitting was performed using the Doniach and Šunjić line shape.<sup>37</sup> The best fitting results are shown in Fig.  $4(a)$  by solid curves. Each fitting result consists of three spin-orbit split curves; the two asymmetric curves correspond to the peaks due to Ba in the  $Si<sub>24</sub>$  and  $Si<sub>20</sub>$  cages, as already shown by A and B, respectively, and the other is the broad peak with smaller amount, which can be considered a surface component. The conclusive fitting curves match well with the photoemission spectra. The asymmetric parameters of peak A are estimated to be 0.375, 0.250, 0.130, and 0.080 for *x*=0, 1, 3, and 6 by the curve fitting, respectively. The systematic transformation of the Ba 4*d* spectra from asymmetric shape for  $Ba_8Si_{46}$  into more symmetric shape for  $Ba_8Ag_6Si_{40}$  is consistent with the changes in the DOS at  $E_F$  and the  $T_c$  depending on Ag content.

Figure 4(b) shows the Si 2*p* spectra of Ba<sub>8</sub>Ag<sub>*x*</sub>Si<sub>46−*x*</sub>  $(x=0,1,3,6)$ . Although the Si 2*p* spectra are complicated in Ag-doped clathrates, a curve fitting was attempted to understand the spectral features. In  $Ba_8Si_{46}$ , there are three unequal sites, which are namely 24*k*, 16*i*, and 6*c* sites. The 6*c* site is the site which is represented by black circles in Fig. 1 and the 24*k* site is adjacent to the 6*c* site. The second nearest neighbor to the 6*c* site is the 16*i* site. In spite of the presence of these different Si sites, the Si  $2p$  spectrum of  $Ba_8Si_{46}$ shows one bulk component with large intensity and small contributions located at the low and high binding energy sides to the main peak [Fig.  $4(b)$ ]. The only one large component in Si 2*p* spectrum indicates that the charge is homogeneously distributed on Si network in  $Ba_8Si_{46}$ , in contrast to the difference between the valences of Ba in the  $Si<sub>24</sub>$  and  $Si<sub>20</sub>$ cages. The fitting spectra for Ag-doped Si clathrates show two main components. Hence, Ag doping causes the Si atoms with two different valences. In Fig. 4(b) the main component at the high binding energy side is marked with A and that at the low binding energy side with B. In  $Ba_8Ag_1Si_{45}$ , peak A is located at higher binding energy than the main peak of  $Ba_8Si_{46}$ . The bonding between Ag and Si is expected to have the larger electron density at the Ag site because the electronegativity of Ag is still greater than that of  $Si^{33}$ Hence, the Si 2p spectrum of  $Ba_8Ag_1Si_{45}$  shows the component which shifts to the higher binding energy. The peak shift of the Si  $2p$  spectrum in Ba<sub>8</sub>Ag<sub>1</sub>Si<sub>45</sub> is consistent with the valence band spectrum of  $Ba_8Ag_1Si_{45}$ , which shows the drastic decrease in the *p*DOS of Si. In  $x \ge 3$ , on the other hand, the component (B), which greatly shifts to low binding energy, is observed in the Si 2*p* spectra. Since Ba provides the electrons for the Si network at the  $Si<sub>24</sub>$  cage through doped Ag in  $x \ge 3$  as was shown in the peak shift (A) of the Ba 4*d* spectra, the excess electron charge on Ag possibly causes conversion of the bonding character so as to provide the electrons for the Si network through Ag in  $x \ge 3$ . The provided electrons to Si sites are partially used for the hybridization with Ba at the  $Si<sub>20</sub>$  cage as was shown by the shift of peak B in Ba 4*d* spectra and are probably also added to the bonding states at  $\sim$ 1.5 eV in the valence band. Hence, complicated charge balance depending on Ag content results so as to stabilize  $Ba_8Ag_xSi_{46-x}$  energetically. In  $Ba_8Ag_6Si_{40}$ , since all of the 6*c* sites are replaced with Ag atoms, the Si 2*p* spectrum is simply assigned by the Si sites. If components A and B are assumed to come from the 24*k* and 16*i* sites, the ratio between A and B is expected to be 0.6:0.4, which approximately matches with the intensity ratio 0.52:0.48 estimated by the curve fit in  $Ba_8Ag_6Si_{40}$ .

In conclusion, valence band and core-level photoemission using soft x rays have been applied to  $Ba_8Ag_xSi_{46-x}$  $(x=0, 1, 3, 6)$ . In the valence band spectra of Ba<sub>8</sub>Si<sub>46</sub>, a sharp peak was prominently observed at  $E_F$ . The resonant photoemission confirmed the strong hybridization with the Ba 5*d* state for this peak. The peak intensity at  $E_F$  decreases with increasing Ag content. In  $Ba_8Ag_1Si_{45}$ , the valence electron of Si is attracted to Ag, which causes the decrease in the DOS at *EF*. The resonant photoemission and Ba 4*d* core-level spectrum have shown that the Ba 6*s* state admixes in the valence band of  $Ba_8Ag_1Si_{45}$ . In  $x \ge 3$ , conversion of the bonding character takes place. In contrast to  $Ba<sub>8</sub>Ag<sub>1</sub>Si<sub>45</sub>$ , the 5*d* electron of Ba inside the  $Si<sub>24</sub>$  cage is exclusively attracted to Ag in  $x \geq 3$  and the DOS decreases further as the result that the hybridization with the Ba 5*d* state weakens. The excess electron charge on Ag flows to Si network. The electron-flow to Si network would form the additional bonding states at  $\sim$ 1.5 eV with doped Ag. As the result of these interchanges of the charge between the elements of Ba<sub>8</sub>Ag<sub>*x*</sub>Si<sub>46−*x*</sub>, Ag doping causes the decrease in the peak intensity at  $E_F$ , which is the dominant factor decreasing the  $T_c$ . The absence of superconductivity in  $Ba_8Ag_6Si_{40}$  for which  $N(E_F) \neq 0$  indicates the weak vibrational coupling between Ba and cages consisting of Si and Ag in  $Ba_8Ag_6Si_{40}$ . The change of the valence band by Ag doping was consistent with the LDA calculation.  $Ba_{8-x}Si_{46}$  (Refs. 6 and 7) and  $Na_{x}Ba_{8-x}Si_{46}$  (Refs. 2 and 3) in which Ba atoms are deintercalated and partially replaced by Na atoms, respectively, and Ba<sub>8</sub>Si<sub>46−*x*</sub>Ge<sub>*x*</sub> (Ref. 38) in which Si atoms are partially replaced with Ge atoms also show the variation of  $T_c$ . Therefore, the experimental studies of the electronic states for these systems are desirable for understanding the overall mechanism on the appearance of the narrow peak at  $E_F$  and superconductivity in Si clathrate compounds.

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