Indication of intrinsic room-temperature ferromagnetism in $Ti_{1-x}Co_xO_{2-\delta}$ thin film: An x-ray magnetic circular dichroism study

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Soft x-ray magnetic circular dichroism (XMCD) measurements at the Co $L_{2,3}$ edges of Co-doped rutile TiO₂ at room temperature revealed clear multiplet features characteristic of ferromagnetic Co²⁺ ions coordinated by O²⁻ ions, being in sharp contrast to the featureless XMCD spectrum of Co metal or metallic clusters. The absorption and XMCD spectra agree well with a full atomic-multiplet calculation for the Co²⁺ high-spin state in the D_{2h} -symmetry crystal field at the Ti site in rutile TiO₂. The results indicate that the ferromagnetism arises from the Co²⁺ ions substituting the Ti⁴⁺ ions. © 2006 American Institute of Physics. [DOI: 10.1063/1.2236829]

Dilute magnetic semiconductors with room-temperature ferromagnetism are promising materials for spintronics devices, in which both spin and charge degrees of freedom are utilized.^{1,2} The finding of ferromagnetism at room temperature in Co-doped rutile and anatase TiO₂ (Refs. 3 and 4) triggered subsequent intensive studies of various oxide semiconductors doped with magnetic ions.^{5,6} Shinde et al. studied ferromagnetic nanoparticles that vanish with hightemperature annealing and on intrinsic ferromagnetism remaining after a heat treatment in anatase-type $Ti_{1-x}Co_xO_{2-\delta}$ films.⁷ Measurements of the anomalous Hall effect⁸ (AHE) and visible-UV magnetic circular dichroism⁹ (MCD) in rutile-type $Ti_{1-x}Co_xO_{2-\delta}$ indicated that charge carriers are responsible for intrinsic ferromagnetism. However, the claim for intrinsic ferromagnetism by AHE was disputed by subsequent work,10 suspecting that Co nanoparticles may be responsible for the AHE. Thus, whether the ferromagnetism is intrinsic or extrinsic has been the subject of much debate.

Soft x-ray magnetic circular dichroism (XMCD) in corelevel absorption is especially suitable for approaching the present issue, because it provides element-specific, valenceorbital-selective, microscopic information about the electronic and magnetic states. XMCD measurements were reported on, e.g., Mn-doped GaAs (Ref. 11) and on Co-doped ZnO,¹² showing ferromagnetism at low temperatures. Shimizu *et al.*¹³ measured Co *K*-edge x-ray absorption spectra (XAS) in anatase-type Co-doped TiO₂ films, confirming the valence state of Co as 2⁺, and detected an XMCD signal, which probes the Co 4*p* states directly, but not the magnetic Co 3*d* states. Kim *et al.*¹⁴ have studied Co-doped anatasetype Ti_{1-x}Co_xO₂ thin films with various *x* by XMCD measurements at the Co $L_{2,3}$ edges. On the contrary, they observed an XMCD spectral line shape nearly identical to that of metallic Co, and found that this XMCD signal increased with annealing the sample. It was therefore suspected that the ferromagnetism in Ti_{1-x}Co_xO₂ was due to segregated metallic Co clusters.¹⁴ Thus, it is still open whether the reported room-temperature ferromagnetism is intrinsic or not.

Here, we report on a combined experimental and theoretical Co $L_{2,3}$ -edge XMCD study of rutile-type Ti_{0.97}Co_{0.03}O_{2- δ} as-deposited films. We observed clear multiplet features at the Co $L_{2,3}$ edges in the XMCD spectrum corresponding to those in XAS of Ti_{0.97}Co_{0.03}O_{2- δ}. The experimentally observed XMCD multiplet features agree qualitatively well with the results of a full atomic-multiplet calculation for high-spin Co²⁺ ions under D_{2h} -symmetry crystal field around the Ti site in rutile TiO₂. Our experimental and theoretical observations strongly indicate intrinsic ferromagnetism arising from Co²⁺ ions substituting the Ti⁴⁺ ions.

Rutile-type $Ti_{0.97}Co_{0.03}O_{2-\delta}$ epitaxial films were synthesized by the pulsed laser deposition method at 400 °C at oxygen pressure of 1×10^{-7} Torr. Ferromagnetism was confirmed at room temperature for the present films by both anomalous Hall effect and magnetization measurements.⁸

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FIG. 1. (Color online) Co L2,3-edge XAS and XMCD of rutile-type $Ti_{0.97}Co_{0.03}O_{2-\delta}$ (a) Photon helicity (h) dependent XAS recorded at 300 K and $B = \pm 1$ T. μ_+ and μ_- stand for the absorption coefficients with h parallel $(\uparrow\uparrow)$ and antiparallel $(\uparrow\downarrow)$ to the Co 3*d* majority-spin direction. An energy integral, r, gives $A_{L_3}+A_{L_3}$, defined in the text. Arrows A–G denote multiplet features. (b) XMCD spectrum and its energy integral. Integrals p and qrepresent ΔA_{L_2} and $\Delta A_{L_2} + \Delta A_{L_2}$, respectively, in the text.

The present samples also showed a MCD signal in the visible-UV region at room temperature.9 XAS and XMCD spectra at the Co L2,3 edges were measured on bendingmagnet (BL-11A) and helical-undulator (AR-NE1B) beamlines at the Photon Factory. Polarization-dependent high resolution XAS were measured at room temperature with the total electron-yield method without Ar ion sputtering or annealing in order to avoid the segregation of metallic Co nanoclusters induced by surface treatments.¹⁴ Magnetic fields of ± 1 T were applied to the samples.

Figure 1 shows the Co $L_{2,3}$ -edge XAS and XMCD spectra of the as-deposited rutile-type $Ti_{0.97}Co_{0.03}O_{2-\delta}$ film. Here, μ_+ and μ_- stand for the absorption coefficients for the photon helicity, h, parallel and antiparallel to the Co 3d majorityspin direction, respectively. Both the XAS and XMCD ($\Delta \mu$ $=\mu_{+}-\mu_{-}$) spectra were corrected for the degree of circular polarization of the incident light. The XAS of the rutile-type $Ti_{0.97}Co_{0.03}O_{2-\delta}$ thin film showed multiplet features. In the following, we refer to each multiplet feature as A-G. The XMCD spectrum showed clear multiplet features that almost corresponded one-to-one to those in the XAS.

Figure 2 shows the expanded XAS and XMCD spectra of $Ti_{0.97}Co_{0.03}O_{2-\delta}$ in the Co L_3 -edge region. The XMCD spectrum of metallic Co is also shown in Fig. 2 for a comparison. A striking point here is that the XMCD spectrum of $Ti_{0.97}Co_{0.03}O_{2-\delta}$ showed a clear negative peak at the energy corresponding to feature D in the XAS. It is emphasized that this negative XMCD peak at D was not observed in the XMCD spectrum of metallic Co. The dominant negative peak in the XMCD spectrum showed a line shape more flat-Downloaded 15 Jun 2007 to 133.41.149.126. Redistribution subject to AIP license or copyright, see http://apl.aip.org/apl/copyright.jsp

FIG. 2. (Color online) Enlarged plots of spectra in the Co L_3 region of $Ti_{0.97}Co_{0.03}O_{2-\delta}$ (a) Co L_3 -edge XAS. (b) Co L_3 -edge XMCD. Note the multiplet features denoted by arrows in the XMCD spectrum, corresponding to features A-E in the XAS. This peculiar XMCD spectrum is in sharp contrast to the smooth and featureless XMCD spectrum of metallic Co, which is plotted for a comparison on a scale reduced by a factor of 0.06.

tened than that in the XMCD spectrum of metallic Co, indicating overlapping, unresolved multiplet features corresponding to features B and C in the XAS. The XMCD spectrum showed a positive peak corresponding to feature A in the XAS. No corresponding feature was seen in the XMCD spectrum of metallic Co. Similarly, the XAS and the XMCD spectra at the Co L_2 edge showed corresponding multiplet structures F and G (Fig. 1). The present XMCD spectra of $Ti_{0.97}Co_{0.03}O_{2-\delta}$ are distinctly different from the previous results by Kim et al.,14 where they studied $Ti_{1-x}Co_xO_2$ samples heat treated prior to the measurements. They observed no multiplet features at the Co $L_{2,3}$ edges in their XMCD spectra, attributing the XMCD to segregated metallic Co clusters. The segregated Co clusters in their samples obviously arose from annealing in a vacuum, as indicated by the systematic increase of XAS and XMCD signals of metallic Co,¹⁴ and may not be due to the intrinsic properties of $Ti_{1-x}Co_xO_2$. In contrast, the present experiment clearly revealed multiplet features in the XMCD spectrum corresponding to those in the XAS without any annealing, verifying intrinsic ferromagnetism arising from Co²⁺ ions that substitute Ti^{4+} ions and are coordinated by O^{2-} ions.

We now compare in Fig. 3 the experimental XAS and XMCD spectra with the results of full atomic-multiplet calculations. The calculation method is described elsewhere.¹⁵ The calculations were made for a low-spin Co2+ ion in a crystal field with O_h symmetry, the high-spin Co²⁺ ion in a crystal field with O_h and D_{2h} symmetries. D_{2h} is the local symmetry around the Ti site in the rutile-type structure. The possibility of the low-spin Co²⁺ ion is immediately excluded from a clear disagreement of the experimental XAS and



FIG. 3. (Color online) Comparison of the experimental spectra with atomicmultiplet calculations. (a) XAS. (b) XMCD spectra. Calculations were done for the low-spin Co^{2+} ion and the high-spin Co^{2+} ion in crystal fields with O_h symmetry, and the high-spin Co^{2+} ion in a crystal field with D_{2h} symmetry. The calculated XMCD spectra were scaled by a factor of 1/30 for a comparison with the experimental XMCD spectrum.

XMCD in line shape with the theoretical ones. The feature corresponding to D at the L_3 edge in the XAS is theoretically reproduced for the high-spin Co²⁺ ion under both O_h and D_{2h} symmetries. Although the overall XMCD spectral line shape is similar between O_h and D_{2h} symmetries, the two features F and G at the L_2 edge (Fig. 1) in both the theoretical XAS and XMCD, and the intensity ratio of the theoretical XMCD at the L_2 edge to that at the L_3 edge show better agreement with the experiment for D_{2h} than for O_h . The experimental XAS and XMCD spectra thus show qualitatively the best agreement with the calculated spectra for the Co²⁺ high-spin configuration in the D_{2h} crystal field. This result further supports that ferromagnetic Co ions have the Co²⁺ high-spin electron configuration and are located at the Ti sites of TiO².

We evaluated the orbital magnetic moment, $m_{\rm orb}({\rm Co})$, and the spin magnetic moment, $m_{\rm spin}({\rm Co})$, of the Co²⁺ ion using the XMCD sum rules,^{16,17}

$$m_{\rm orb} = -\frac{4}{3} \frac{\Delta A_{L_3} + \Delta A_{L_2}}{A_{L_3} + A_{L_2}} \cdot n_h \mu_B, \tag{1}$$

$$m_{\rm spin} + 7m_T = -\frac{2[\Delta A_{L_3} - 2\Delta A_{L_2}]}{A_{L_3} + A_{L_2}} \cdot n_h \mu_B, \qquad (2)$$

where A_{L_2} (A_{L_3}) and ΔA_{L_2} (ΔA_{L_3}) are the L_2 - (L_3 -) edge integrated XAS and XMCD intensities, respectively, n_h is the 3*d* hole number, and m_T is the magnetic quadrapole moment. By using the integrals¹⁸ p, q, and r in Fig. 1, and assuming a nominal value of n_h =3.0 in a Co²⁺ ion, the orbital and the

effective spin moments were determined to be $m_{orb}(Co) = (0.013 \pm 0.002)\mu_B$ and $m_{spin}(Co) + 7m_T(Co) = (0.12 \pm 0.01)\mu_B$. We thus obtained the total moment $m_{total}(Co) = m_{spin} + m_{orb} + 7m_T = (0.13 \pm 0.01)\mu_B$. The value of $m_{total} = 0.13\mu_B$ is smaller by a factor of $\sim 7-8$ than $m_{total}(Co) = 1.0\mu_B$, determined by magnetization measurements.⁸ The origin of this discrepancy is not clear at present, but a possible cause is the formation of a magnetically dead layer on the film surface.

In conclusion, we have verified that the ferromagnetism in a rutile-type $Ti_{0.97}Co_{0.03}O_{2-\delta}$ film is caused by high-spin Co^{2+} ions substituting the Ti^{4+} ions on the basis of an element-specific XMCD study. The present result provides key information for understanding of dilute magnetic semiconductors based on wide band-gap oxides and for the development of spintronic devices using them.

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